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Analysis of the Formed Protective Layer Inhibiting Alkali Corrosion in Aluminosilicate Refractory Castables

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Abstract: This article analyzes the mechanism of the protective layer formation under the action of alkali in a refractory castable when ground quartz sand (GQS) is used as an admixture to produce refractory conventional castables (CC) and medium cement castables (MCC). It was found that, independently of the castable composition, the addition of GQS (2.5%) reduces the degree of K_2CO_3 dissolution at high temperature, and the released potassium reacts with the silica and forms a viscous potassium silicate glass, which reduces the mobility of alkali. The liquid phase formed filled some of the open pores and hindered the penetration of potassium into the deeper layers of the refractory castable. The thickness of the formed protective layer, after three cycles of the alkaline corrosion test, varies from 700 μm up to 1300 μm, depending on the castable composition.

Keywords: refractory castable; aluminosilicate; ground quartz sand; alkali resistance; protective layer

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1. Introduction

Various aggressive alkaline compounds, such as potassium and sodium carbonates, chlorides, and sulphates are formed in the biomass combustion process [1], produce ash of low melting temperature, and pose a serious challenge for the application of biomass in energetic plants instead of fossil fuels. This leads to a more intensive fouling of the boilers, deposition of fly ash on the heating surfaces, and alkali corrosion of materials used in the equipment [2,3]. Corrosion of refractory materials significantly reduces the durability of the lining [4,5] of the boilers and also causes failure [6].

In order to prevent premature degradation of the material due to ash's influence, boilers' operating parameters and requirements for biofuel usually are quite strict. Therefore, it is expected that there are no conditions for significant alkali attack. Unfortunately, various violations (poorly controlled local temperatures, the use of low-quality fuel with an increased content of alkalis and other aggressive impurities, the use of alkaline salts to prevent freezing of biofuels, etc.) often lead to emergency situations associated with the rapid destruction of the internal structures of the biomass boilers (Figure 1).

The reaction of aluminosilicates, which are present in refractory materials used for biomass boilers (fireclay, mullite, corundum), with potassium compounds produces different corrosion products, such as kaliophilite ($K_2O{\cdot}Al_2O_3{\cdot}2SiO_2$), leucite ($K_2O{\cdot}Al_2O_3{\cdot}4SiO_2$), and other compounds with the volume exceeding the initial volume of the material [7–10].

The volume of reaction products in fireclay can increase by 19% and by 22% in mullite with approx. 32% of K_2O penetrated [8]. The 20–25% expansion of mullite and corundum leads to the degradation of the material [9].

Alkali metal ions penetrate most easily into the matrix of material [11], accumulate in the matrix, and only then react with the aggregates [12]. This process takes place due to the higher porosity of the matrix compared to the aggregate and usually a higher content of various impurities (Na_2O, TiO_2, and other compounds) [13]. After the penetration into the aluminosilicate material, the alkali metals react first with the glass phase of the material or

cristobalite and only later with the mullite [14]. The authors note that the reaction between K_2O and SiO_2 produces potassium silicate, which reacts with mullite to form kaliophilite over a short time period, followed by a longer reaction of the latter with the free SiO_2 to form leucite.

Figure 1. Decomposition of corroded biomass boiler lining.

The penetration of alkali vapor or alkaline liquid phase into the structure of a material depends on its open porosity and other porosity parameters [15]. Corrosion is less likely to develop in materials with lower porosity and gas permeability [13,16,17].

The potential of crystalline SiO_2 should be considered to reduce alkali penetration into the aluminosilicate structure by adjusting the chemical composition of the refractory material. Silica-enriched melt from the K_2CO_3-$CaCO_3$-SiO_2 mixture (synthetic ash) was found [12] to significantly hinder the penetration of K into mullite (63% Al_2O_3) and corundum (83% Al_2O_3)-based material, compared to the pure K_2CO_3 or K_2CO_3-$CaCO_3$ mixture used in the tests described in the paper. Another paper reports [18] that the refractory containing 35% of Al_2O_3 had higher resistance to alkali due to the higher content of free silica in its matrix, compared to the refractory containing 50% of Al_2O_3. The alkali and silica reaction produces a viscous glass phase that prevents the alkali from penetrating deeper into the refractory material [19]. Our previous research into the alkali resistance of concrete [20,21] showed that the addition of finely ground quartz sand to fireclay castable promotes the formation of a protective glassy layer on the surface of the castable exposed to K_2CO_3 at 1100°C. The resistance of fireclay castables to alkali attack determined by the static crucible method increased by more than five times (from 4 to more than 20 cycles).

However, the mechanism of the formation of a protective layer in aluminosilicate castables (subject to the chemical composition and porosity parameters of the material) modified with finely ground quartz sand is not sufficiently clear.

The aim of this work is to investigate the formation of a protective layer that inhibits alkaline corrosion in different types of refractory castables with fireclay aggregate and modified with quartz sand at 1100 °C temperature.

2. Materials and Methods

Conventional (cement content 25%, CC) and medium cement (12%, MCC) refractory castables with different types of fireclay aggregate (BOS135 and BOS145, particle size less than 4 mm), produced by Tabex-Ozmo (Poland) with Al_2O_3 content in BOS135 ~35%, Al_2O_3/SiO_2 (A/S) ratio 0.68, and Al_2O_3 content in BOS145~45%, with A/S ratio 0.89 were tested. The castables were labelled as CC-1, CC-2, MCC-1, and MCC-2, respectively. Calcium aluminate cement Istra 40 (CAC40) produced by Calucem GmbH (Mannheim, Germany) was used for CC mixes, and Gorkal 70 (CAC70) cement produced by Gorka Cement Company (Trzebinia, Poland) was used for MCC. The castable also contained ultra-dispersive admixtures such as microsilica (MS) RW-Fuller produced by RW Silicon GmbH (Pocking, Germany), reactive alumina (RA) CTC 20 and calcined alumina (CA) CT 19 produced by Almatis (Ludwigshafen, Germany), and deflocculants (Castament FS20,

Castament FS30, and sodium tripolyphosphate (NT)) produced by BASF Construction Solutions GmbH (Trostberg, Germany). Ground quartz sand (GQS) containing 99.20 % SiO_2 (average particle size about 10 μm) produced by AB Anykščių kvarcas (Anykščiai, Lithuania) was used to induce the development of the protective layer. The compositions and the main properties of the castables are given in Table 1. The 2.5% of GQS was chosen, because this amount significantly increases the alkali resistance of different types of castables, and thermal shock resistance decreases insignificantly. A larger amount of GQS (up to 7.5%) decreases thermal shock resistance of castables depending on the castable type by up to 70% [22].

Table 1. Compositions of castables (mass %) and their characteristics [23].

Mark of Composition	Content of the Components											Characteristics	
	CAC70	CAC40	MS	RA + CA	BOS135	BOS145	GQS	FS20 *	NT *	FS30 *	Water *	Compressive Strength, MPa	TSR, Cycles
CC-1		25	2.5	-	70.0	-	2.5	-	-	0.1	10	51	10
CC-2	-	25	2.5	-	-	70.0	2.5	-	-	0.1	8.5	67	17
MCC-1	12	-	5.0	12	68.5	-	2.5	0.1	0.1	-	9.0	76	24
MCC-2	12	-	5.0	12	-	68.5	2.5	0.1	0.1	-	8.0	126	29

*—over 100 % of the key dry substances.

The results (Table 1) on the compressive strength for the CC and MCC series castables with GQS after firing at 1100 °C and on the thermal shock resistance (TSR) determined at 950 °C show that different classes of castables were used in the alkaline corrosion tests. Castables with different properties were chosen for analysis to determine how different mechanisms of protective layer formation could occur during the impact of potassium at 1100 °C temperature.

The specimens of refractory castable were prepared, dried, and fired according to the requirements set out in LST EN ISO 1927-5:2013. The physical mechanical characteristics were established according to LST EN ISO 1927-6:2013, deformation temperatures were measured according to EN ISO 1893:2009, and thermal shock resistance was determined according to DIN 51068. The open porosity of castable was determined according to LST EN ISO 10545–3:2000.

The chemical composition of the refractory castable was analyzed by means of wavelength dispersive X-ray fluorescence spectroscopy (XRF-WD). The analysis was performed on ZSX Primus IV spectrometer (Rigaku Corp., Tokyo, Japan) 4 kW, Rh anode.

The crucible method (ASTM C 454-83:2007) was chosen to assess the influence of alkali attack on the structure of the castable [23]. For the tests, 70 × 70 × 70 mm castable specimens with a cylinder-shaped cavity of a diameter of 20 mm and a height of 30 mm (4 specimens for a series of castable) were made. The specimens were dried at the temperature of 110°C and fired at the temperature of 1100 °C. After adding 9 g K_2CO_3 into the cylinder-shaped cavity, the specimens were fired at the temperature of 1100 °C for 5 h. The cavity was covered with another specimen (70 × 70 × 20 mm). After multiple tests (upon adding 9 g K_2CO_3 for each test), the specimens were inspected visually and an appearance of microcracks in them was registered. For SEM/EDS analysis of the affected zones of the castable, a specimen of a composition identical to the specimen used for covering the cavity of the specimen was used (Figure 2, point 2). After three cycles of the test, a specimen was cut out from the said specimen; the surface of the latter was strengthened by resin and then

smoothed and polished. For X-ray and mercury intrusion porosity after 3 alkali resistance cycle tests, the samples were cut from the same point 2 area (Figure 2).

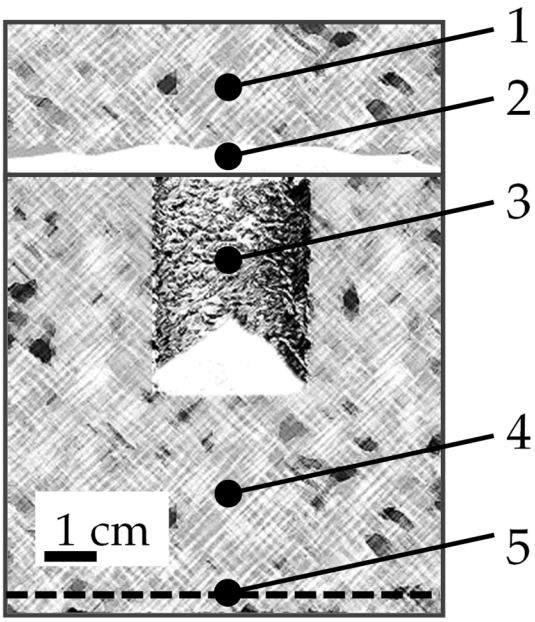

Figure 2. The position of test specimens: 1—plate is used for SEM and EDS; 2—a protective layer had formed on the surface of the plate; 3—cavity with K_2CO_3 salt; 4—the crucible method test sample; 5—unreacted castable layer was used for XRD analysis.

A scanning electron microscopy device (SEM JEOL JSM-7600F) was used (Tokyo, Japan) for the tests. The parameters of the scanning electron microscope were as follows: voltage 10 kV, the distance to the surface of the specimen 7 to 10 mm. X-ray microanalysis was performed by the energy dispersion spectrometer (EDS) Inca Energy 350 (Oxford Instruments, Oxford, United Kingdom), using Silicon Drift type detector X-Max20. The INCA Energy software package (Oxford Instruments, Oxford, United Kingdom) was used.

For X-ray tests, the sample of the castable was crushed and sieved (sieve opening size 0.063 μm). X-ray diffraction (XRD) analysis of powder was conducted with diffractometer DRON-7 (St. Peterburg, Russia) with Cu-Kα (λ = 0.1541837 nm) radiation. The following settings were used: 30 kV voltage; 12 mA current; 2θ diffraction angle range from 4° to 60° with increment of 0.02° measured each 2 s. The phases were identified comparing the XRD images with standard diffraction patterns provided by the International Centre for Diffraction Data (ICDD).

Mercury intrusion porosimetry (MIP) method was applied to analyze the parameters of castable pore structure by means of Quantachrome Poremaster 33/60 (Quantachrome Instruments, Boynton Beach, Florida, USA) with a maximum pressure of 33,000 psi for pore diameter ranging from 1100 μm to 0.0035 μm and with two low-pressure stations plus one high-pressure station.

The interaction between the K_2CO_3 salt and the castable was tested by performing a thermal analysis curve of the mix K_2O/castable = 1/9. Additionally, samples without GQS (CC-1′, CC-2′, MCC-1′, MCC-2′) were prepared to compare differences in mass loss at temperature up to 1000 °C. The amount of GQS was then replaced by milled chamotte BOS135 and BOS145 according to the mix (the specific surface area of milled BOS135 is 409 m^2/kg and for BOS145 395 m^2/kg.). Thermal analysis (DTG) was performed with a Perkin Elmer TGA 4000 (Waltham, Massachusetts, USA) thermal analyzer. The samples of the mixture with a mass of 40–50 mg were placed in a platinum crucible and heated to 1000 °C at 10 °C/min in a nitrogen environment.

3. Results

3.1. Chemical and Mineral Composition of Castables

Chemical and mineral compositions of the tested castables was determined for further evaluation of the interaction between potassium and aluminosilicate material (Table 2 and Figure 3).

Table 2. Chemical composition of the tested castables, mass %.

No.	Chemical Composition, Mass %	The Mark of Composition			
		CC-1	CC-2	MCC-1	MCC-2
1	Al_2O_3	34.47	41.49	44.08	51.10
2	SiO_2	44.55	38.59	46.05	40.52
3	CaO	11.30	11.42	4.64	4.46
4	Fe_2O_3	5.71	5.40	1.97	1.58
5	K_2O	0.92	0.64	0.81	0.54
6	Na_2O	0.13	0.14	0.21	0.25
7	MgO	0.51	0.40	0.41	0.30
8	ZrO_2	0.04	0.04	0.03	0.02
9	TiO_2	1.38	1.38	0.94	0.77
10	P_2O_5	0.07	0.07	0.11	0.11

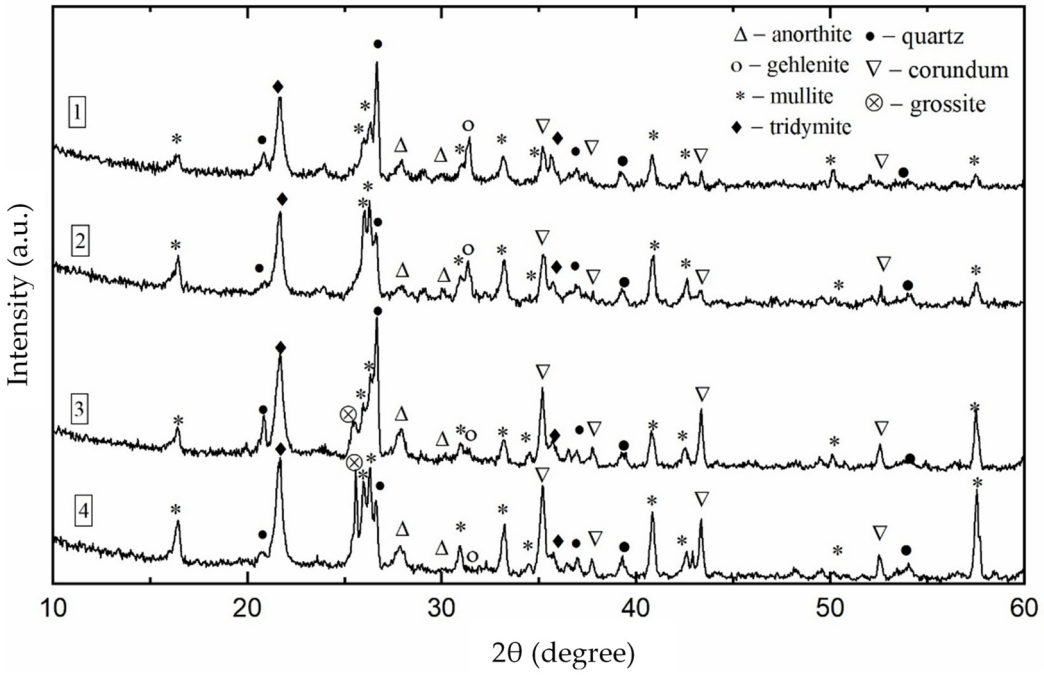

Figure 3. X-ray diffraction patterns of castables after thermal treatment at the temperature of 1100 °C: 1—CC-1; 2—CC-2; 3—MCC-1; 4—MCC-2.

Table 2 illustrates that Al_2O_3 content in the tested castables increases from ~34% to ~51%, and A/S increases from 0.77 to 1.26. CaO content in the MCC series castables is about 2.5 times lower than in the CC series castables due to the lower cement content in MCC compositions. The total content of CaO, Fe_2O_3, MgO, TiO_2, P_2O_5, and alkali oxides reaches ~20% in the CC series castables and ~9.5% in the MCC series castables. It should be noted that these impurities act as a flux for the main components Al_2O_3 and SiO_2 and can promote the formation of the liquid phase at high temperatures [24]; the impurities also affect the maximum temperature of using the aluminosilicate material. The 5% strain point (T_5) was found to be 1290 °C for the CC-1 castable and 1300 °C for the CC-2 castable. The T_5 for the MCC series castables was considerably higher: 1400 °C for MCC-1 and 1480 °C for MCC-2.

Minerals typical of aluminosilicate materials [25], such as mullite, tridymite, quartz, anorthite, gehlenite, and corundum, were identified in all castable samples after thermal treatment at a temperature of 1100 °C (Figure 3). Grossite was identified in the castables of the MCC series because this mineral is present in CAC70 cement, which was used to produce MCC (Table 1).

3.2. XRD, SEM, and EDS Analyses of the Surface Layer of a Castable Specimen after Three Cycles of Alkali Attack

XRD analysis of the powder from the surface layer of the castable CC and MCC series (Figure 4) revealed the presence of corrosion products in the crystalline state, i.e., potassium silicate with a low melting point (approx. 765 °C) $K_2Si_4O_9$ (d = 0.271; 0.277; 0.274; 0.211; 0.312 nm), kalsilite $KAlSiO_4$ (d = 0.312; 0.258; 0.397; 0.218; 0.435 nm), and leucite $KAlSi_2O_6$ (d = 0.327; 0.345; 0.541; 0.284; 0.292 nm). It should also be noted that the intensity of quartz peaks was significantly reduced, and anorthite peaks disappeared in the surface layer of a castable specimen after three cycles of alkali attack. Less amount of corrosion products is formed in the MCC-2 castable. The amorphous hump between about 20° and 35° 2θ (particularly pronounced in the curves for MCC-type castables (curves three and four in Figure 4) is indicative of the formation of glass in the surface layer.

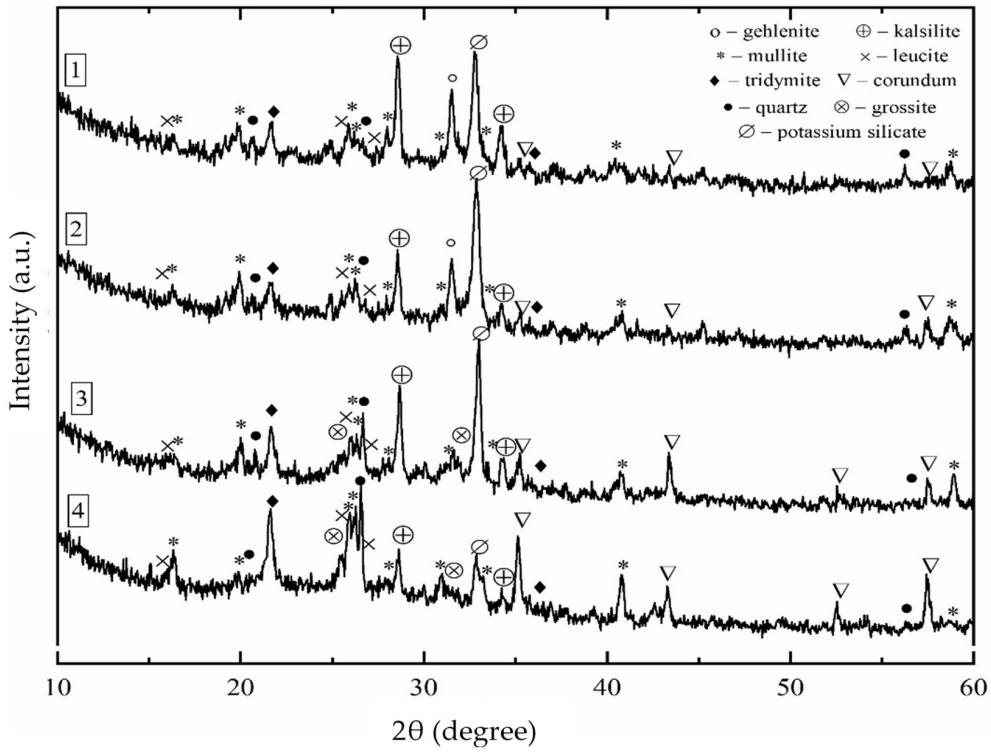

Figure 4. X-ray diffraction patterns: 1—CC-1; 2—CC-2; 3—MCC-1; 4—MCC-2.

Figure 5 illustrates the distribution of potassium obtained by EDS in the analyzed ~1.5 mm thick layer of the castable samples tested after three cycles of alkali attack.

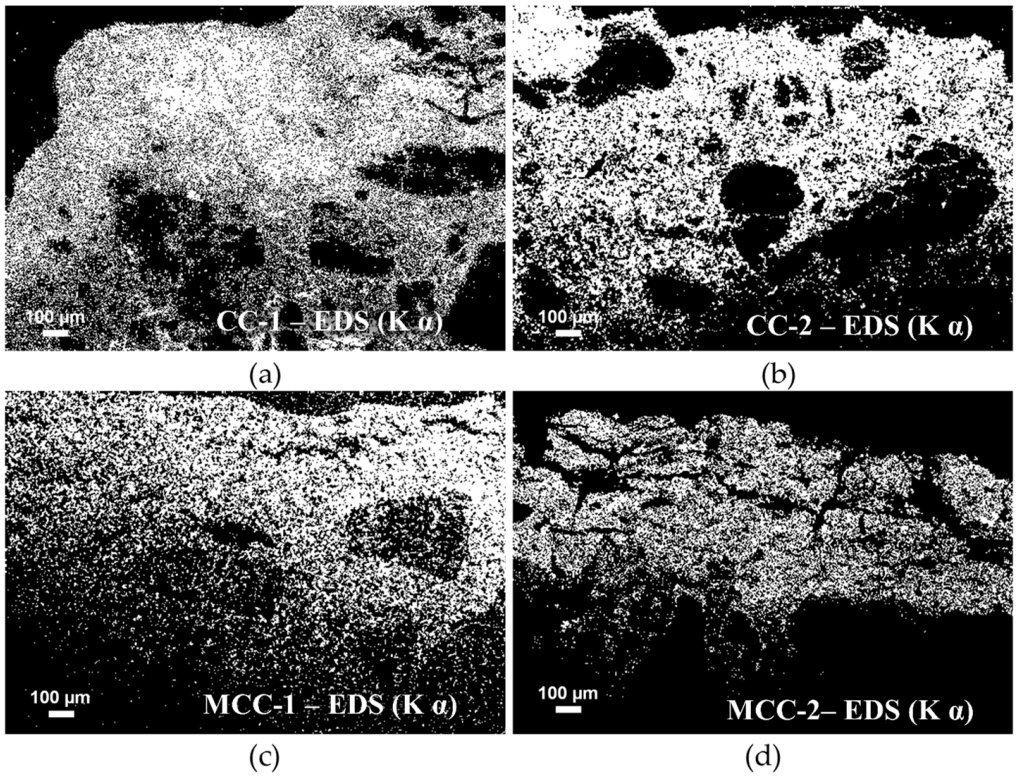

Figure 5. K distribution in the surface layer of castables after 3 cycles of alkali attack at a temperature of 1100 °C seen in EDS maps: (**a**)—CC-1; (**b**)—CC-2; (**c**)—MCC-1; (**d**)—MCC-2.

The concentration of potassium is high in the surface layer of the specimen, where the protective layer has formed, with a clear decrease in the zone below (this is particularly evident in the case of the MCC-2 in Figure 5d). The formation of microcracks and low penetration of K into the aggregate particles can be observed in the protective layer of the MCC.

The detailed distribution of K (100 µm step) obtained by SEM-EDS analysis (the points recorded in the castable matrix) shows (Figure 6) that K is distributed in the protective layer of the tested castables as follows: K concentration in 300–600 µm-thick layer closest to the surface is around 25–35%, whereas the concentration in deeper layers of 1300 µm thickness in CC-1, 1100 µm thickness in CC-2, 1200 µm thickness in MCC-1, and 700 µm thickness in MCC-2 is almost the same, about 21–25%.

Table 3 provides information on the average values of the chemical elements after three cycles of alkaline resistance in the thickness of the protective layer recorded from the EDS analysis (last measurement before the significant decrease in potassium).

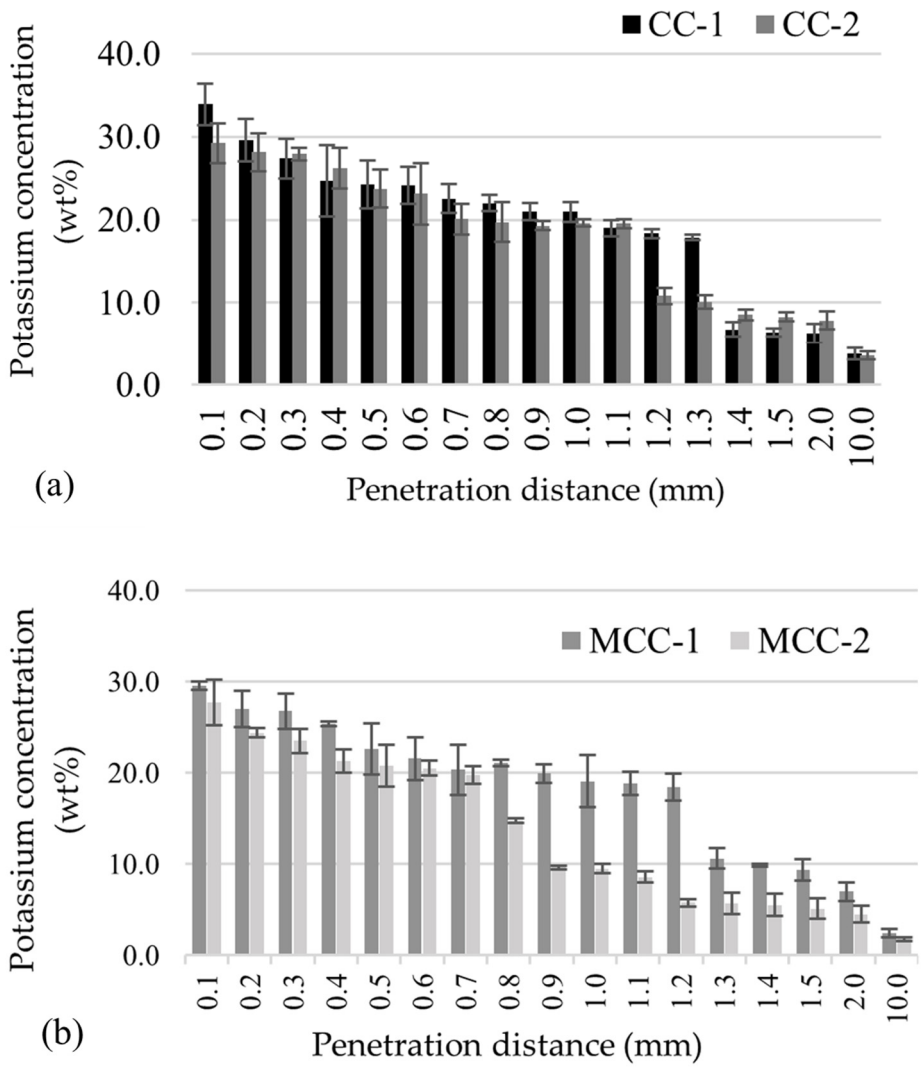

Figure 6. Potassium distribution in the surface layer of castable: (**a**)—CC; (**b**)—MCC.

Table 3. Percentage averages of major chemical elements in the protective layer after 3 cycles of potassium attack.

The Mark of Composition	Thickness of the Layer with High K Concentration	Average Elemental Concentration, %						Total, %
		K	Al	Si	Ca	Na	O	
CC-1	1300 μm	24.5 ± 0.4	14.4 ± 2.2	11.8 ± 1.8	4.1 ± 0.2	0.5 ± 0.04	44.7 ± 2.0	100.0
CC-2	1100 μm	21.5 ± 0.9	16.6 ± 1.8	11.1 ± 2.0	3.8 ± 0.3	0.4 ± 0.05	46.6 ± 1.5	100.0
MCC-1	1200 μm	24.3 ± 0.6	18.9 ± 0.9	14.1 ± 0.3	1.4 ± 0.1	0.7 ± 0.1	40.6 ± 2.1	100.0
MCC-2	700 μm	22.8 ± 0.7	19.5 ± 0.5	13.8 ± 0.5	0.9 ± 0.05	0.6 ± 0.1	42.4 ± 2.0	100.0

According to the results, the average elemental concentration reaches 21.5–24.5% for K, 0.9–4.1% for Ca, and 0.4%–0.7% for Na, depending on the composition of the castable. Average concentrations of these elements characterize to a certain extent the glass composition formed in the protective layer and can help determine the viscosity of the melt, which is important in assessing the stability of the layer at high temperatures.

3.3. Porosity Tests

Porosity of the material and its parameters can also influence the parameters of the protective layer in addition to the chemical composition of the material. The determined average open porosity of castables varied depending on the type of castable and the aggregate used: 24.0% and 22.2% for CC-1 and CC-2, respectively, and 20.3% and 19.1% for MCC-1 and MCC-2.

The pore size distribution of all castable compositions was found to follow the Gaussian curve (Figure 7). The maximum pore size goes up to 1 mm. The average pore size of CC-1, CC-2, MCC-1, and MCC-2 not affected by K_2CO_3 was 1.22 µm, 1.38 µm, 0.83 µm, and 0.77 µm, respectively. Figure 7 illustrates that the MCC series not only has a lower open porosity but also has the smallest pore size.

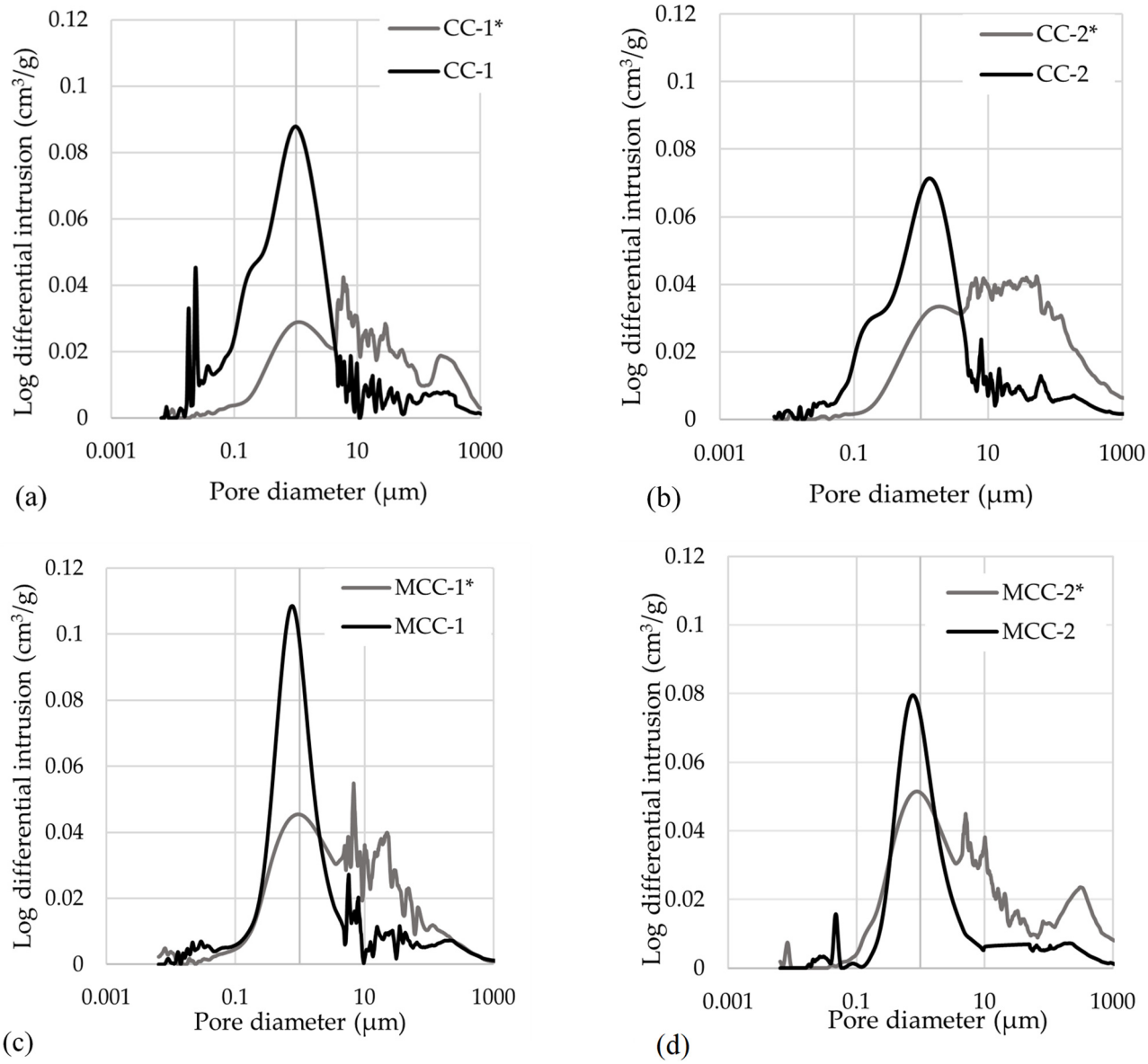

Figure 7. Result of castable porosity before and after 3 cycles of alkaline attack (*): (**a**)—CC-1; (**b**)—CC-2; (**c**)—MCC-1; (**d**)—MCC-2.

Porosity reduced when the refractory castable was exposed to K_2CO_3 (Figure 7, CC-1*, CC-2*, MCC-1*, MCC-2*) as the resulting liquid phase filled in some of the open pores. The maximum log differential intrusion reduced 67%, 53%, 59%, and 35%, respectively. The maximum change in the pore volume was recorded in 1µm pores. The volume of pores smaller than 1 µm in MCC remained unchanged due to the high viscosity of the resulting liquid phase [26].

Higher amounts of Al_2O_3 + SiO_2 (about 27% for CC and 33% for MCC) were found in the cooled glass layer of MCC, leading to increased viscosity of the liquid phase [27] and preventing the filling of the finest pores. The increase in viscosity is strongly influenced by the addition of crystalline SiO_2, which slows down the reaction with K_2CO_3 [28]. The presence of larger pores (>10 µm) or capillaries in the affected castable can increase due to the cycles of alkaline attack, whereby the penetration of potassium can dissolve the surface layer of the castable by forming a structure of interlocking pores and capillaries. The increase in the number of larger pores could also be explained by the devitrification [29] of the corrosion products formed, in particular $K_2Si_4O_9$, by cyclic heating and cooling of the material.

3.4. DTG Test Results of Mix of K_2CO_3 and Castable Powder

DTG analysis was carried out for comparison of the processes that are taking place at high temperatures in the castables with and without GQS. It was determined [20,21] that in samples without GQS, the protective layer does not form.

A broad stage of the mass loss in the temperature range 500–880 °C was observed in the DTG curves of all castable mixtures (Figure 8). It is related to the decomposition of K_2CO_3 and the release of CO_2. Free potassium reacts in the same temperature range with free SiO_2 present in the castables tested to form potassium silicate. Other authors [30–32] report and describe a similar process in different mixtures of materials containing K_2CO_3 and SiO_2 (ash, sludge, pyrophyllite).

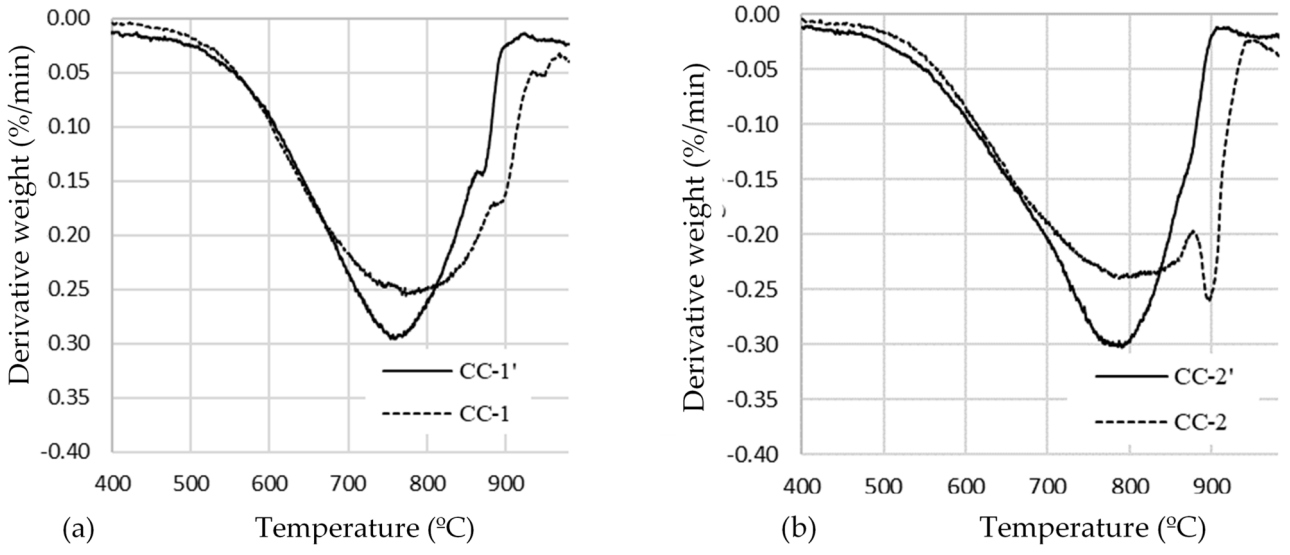

Figure 8. *Cont.*

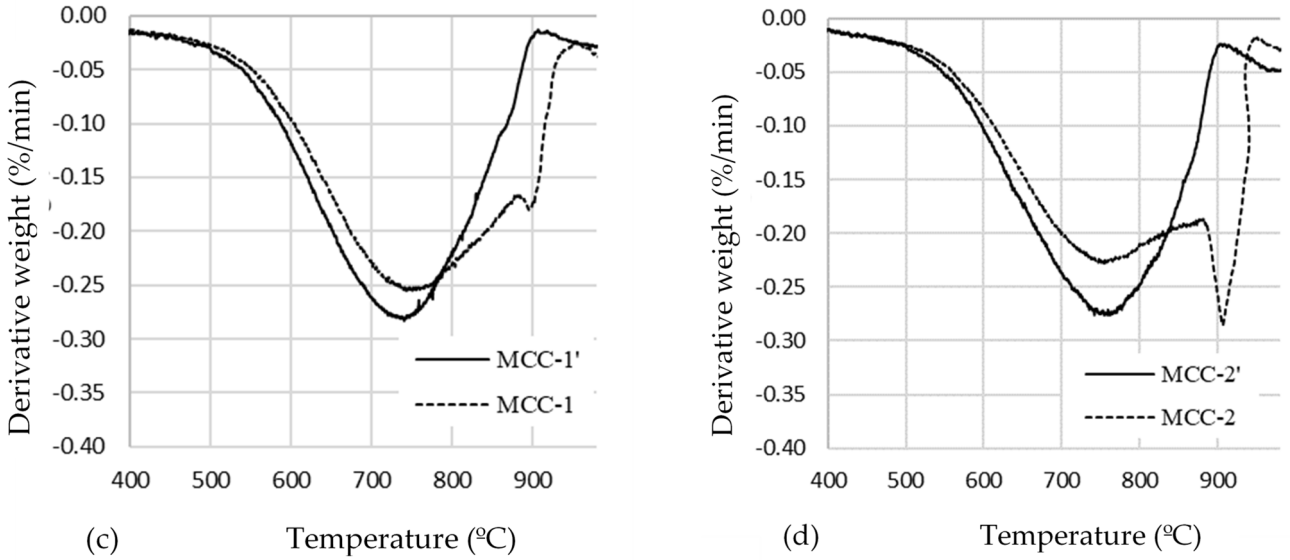

Figure 8. DTG curves of K_2CO_3 and castable powder mixtures: (**a**) a mix without GQS CC-1′ and with GQS CC-1; (**b**) a mix without GQS CC-2′ and with GQS CC-2; (**c**) a mix without GQS MCC-1′ and with GQS MCC-1; (**d**) a mix without GQS MCC-2′ and with GQS MCC-2.

Lower mass loss was recorded in castables with GQS (CC and MCC) in the 500–880 °C temperature range, when the reaction between potassium carbonate and SiO_2 takes place (Table 4, Figure 8) compared to the specimens without GQS (CC′ and MCC′). Moreover, a second stage of the mass loss was observed in the castables with GQS with the peak at 890 °C. This stage is related to the release of CO_2 when the melting point of K_2CO_3 is reached [33]. The mass loss recorded in castable mixtures with a formed protective layer recorded in the temperature range 880–950 °C is 4–5 times higher (Table 4) compared to the mixtures without it (CC′ and MCC′). This stage is particularly pronounced in the CC-2 and MCC-2, which have a lower SiO_2 content (Table 2).

Table 4. The mass loss (%) at different temperature ranges.

Composition	500–880 °C	880–950 °C
CC-1	6.2	0.8
CC-1′	6.3	0.2
CC-2	5.7	1.0
CC-2′	6.3	0.2
MCC-1	5.9	0.8
MCC-1′	6.3	0.2
MCC-2	5.5	1.3
MCC-2′	6.2	0.3

4. Discussion

The liquid phase is formed in the tested aluminosilicate castables without ground quartz sand by the decomposition of the reactant (potassium carbonate) on the surface of the castable specimens and by the reaction of the potassium released with the glass phases in the material [14,34], which are enriched in SiO_2 and Al_2O_3. Such phases form at a temperature below 850 °C [4]. It was established [35] that the viscosity of the molten alkali silicates decreases at 1100–1400 °C when the K concentration increases. In our case, when K concentration during alkali tests increases due to repeated addition of K_2CO_3, the formed liquid phase penetrates easily into the structure of the material, regardless of the open porosity of the castables tested, and mixes with new portions of the glass phases of the material, in a manner similar to the corrosion mechanism described in the paper of [36].

The zones of altered structure develop in the specimens that cause disintegration due to the stresses between the zones of the original and the altered structure [37].

The addition of a relatively small amount (2.5%) of ground quartz sand influences the formation of a protective layer on the surface of the castable and thus makes it difficult for potassium to penetrate, regardless of the castable type, its chemical composition, and porosity. As shown, the characteristics of the tested castables vary over a wide range (Table 1), as well as the open porosity varies from ~19% to ~24% and the chemical composition was also different, taking into account the different A/S and the amounts of various impurities in them (Table 2). It appears that the GQS admixture can be a versatile tool for increasing the alkali resistance of a wide range of aluminosilicate castables.

The additional quartz content in the castable reduces the degree of K_2CO_3 dissolution in the 500–880 °C temperature range, as shown by the reduced mass loss obtained for the dry mixture of CC or MCC powder and K_2CO_3 mixtures (Figure 8). In this case, the released K at the surface of the castable reacts with the silica added to the castable to form a viscous potassium silicate glass [18,35], which effectively reduces the chemical mobility of K. The viscous glass is not able to penetrate the structure of the aluminosilicate material, and the castable specimens withstand alkali attack. Similarly, the authors of paper [36] explain the formation of viscous glass by the arrested penetration of liquid slag Al_2O_3-CaO into the andalusite-based refractory.

The protective layer resulting from the reaction during three cycles of alkaline attack and impeding the penetration of K into the castable structure had a thickness between 700µm and 1300µm, depending on the castable type. The formation of crystalline potassium silicates and potassium alumina silicates was observed in the cooled specimens of all castables with a decrease in the intensity of the quartz peaks and an almost complete disappearance of anorthite.

The distribution of K, Si, and Al in the protective layer indicates that the equilibrium in local scale is achieved in this layer at high temperatures, where the solid phases are in the local equilibrium with the surrounding fluid, similar to the results of the work [36]. Average concentrations of these elements (Table 3) characterize to some extent the glass formed in the protective layer, depending on the type of castable. Al_2O_3, which is set in the protective layer, can influence the viscous glass melting temperature and its viscosity and then, according to the authors [38], the melting temperature and viscosity values tend to increase. Differences in chemical and phase composition in the protective layer affect its stability. Therefore, when operating under constant or cyclic chemical loading, the durability of different castables will be different.

Castable porosity tests after three cycles of alkali attack showed that the liquid phase formed in the protective layer filled some of the open pores. The number of pores smaller than 1µm in the MCC series remained almost unchanged. In the CC series, the pores were filled with the liquid phase, presumably as a result of the dissolution of the pore walls by the liquid at high temperature. This was influenced by the high content of impurities in these castables (Table 2, ~20%).

The formation of microcracks was observed by SEM and MIP in the protective layer followed by high volumetric expansion due to the formation of kalsilite and leucite [15]. This expansion can have an impact on the stability of the protective layer under thermal cycling, and deeper penetration of K is possible as well.

5. Conclusions

The mechanism of the protective layer formation under the action of an alkali in refractory castable when ground quartz sand is used as an admixture to produce refractory conventional and medium cement castables was analyzed.

The characteristics of these castables (A/S, open porosity, etc.) that may influence the penetration of alkalis vary over a fairly wide range. It was found that the addition of ground quartz sand (2.5%) forms a protective layer on the surface of the castable, regardless of the castable type. The thickness of the protective layer after three cycles of the alkaline corrosion test reaches ~700–~1300 μm depending on the type of castable

The amorphous phase in the protective layer increased compared to the castable not exposed to alkali attack, and the following alkali corrosion products were identified in the crystalline state: kalsilite, leucite, and potassium silicate. Potassium silicate and the additional amorphous phase are formed due to the reaction between potassium and quartz sand. The resulting viscous glass fills the pores and capillaries of the material, forming a protective layer which makes it difficult for potassium to penetrate into the deeper layers of the material.

The addition of ground quartz sand reduces the degree of K_2CO_3 dissolution in the 500–880 °C temperature range, and the released potassium at the surface of the castable reacts with the silica and forms a viscous potassium silicate glass, which reduces the mobility of potassium.

Porosity tests after three cycles of the alkaline corrosion test showed that the liquid phase formed in the protective layer filled some of the open pores. The number of pores smaller than 1μm did not change in the MCC, whereas in the CC the pores were filled with the liquid phase.

On the basis of research results, it can be assumed that quartz admixture can be a versatile tool for increasing the alkali resistance of a wide range of aluminosilicate castables.

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